

Numerical Study of Electrically Conductive Graphene/Paraffin Wax Nanocomposites

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Abstract

Over the last years, nano-enhanced Phase Change Materials (nano-PCM) have lately attracted a great attention to address many key barriers (i.e., low electrical and thermal conductivities) to wide adoption of PCMs in emerging applications such as cooling systems, energy conversion, electronic devices, electrodes for rechargeable batteries. For all applications, the knowledge of the electrical properties is important. In this work, a numerical setup was built to investigate the electrothermal behavior of the nanocomposite graphene/paraffin wax. This new materials are developed by associating paraffin wax as thermal storage material and graphene nanoparticles as conductive fillers. The finite element method based on COMSOL Multiphysics 5.2a software, was used to solve the coupled equations of various physical phenomena: electric current, heat transfer and phase change process of the paraffin. To understand the physics-based mechanisms during current distribution, a multi-scale numerical models were developed, where the homogeneous mode is used in macroscopic scale and heterogeneous one in microscopic scale. The choice of optimal physical properties of these materials is analyzed to predict the effective electrical and thermal conductivities of the composite system. The numerical results indicate that the improvement of electrical conductivity of the composite is due to the formation of a conductive network inside the paraffin wax. The resulting current passing through the graphene nanoparticles produces joule heat and self melting of paraffin wax. Accordingly, electric heating behavior of the nanocomposite is strongly dependent on graphene concentrations as well as applied voltage.

Keywords: Numerical study; Graphene nanoparticles; Electrothermal behavior; Nanocomposite materials; Phase change material

Nomenclature

C_p : Heat Capacity ($J \cdot kg^{-1} \cdot K^{-1}$); K : Thermal conductivity ($W \cdot m^{-1} \cdot K^{-1}$); L : Latent heat (J/kg); T : Temperature (K); T_m : Temperature of melting of PCM (K); ΔT : Temperature difference (K); U : Velocity (m/s); σ : Electric conductivity (S/m); ρ : Density ($kg \cdot m^{-3}$); β : Thermal expansion coefficient ($1/K$); μ : Viscosity ($N \cdot s/m^2$); GNP: Graphene nanoparticle; nano-PCM: PCM nanocomposite; PCM: Phase change material

Introduction

Thermal Energy Storage System (TESS) is considered as an especially bright way to minimize the mate between demand and supply of energy. The use of Phase Change Materials (PCMs) seems one of the most significant storage technique due to their ability of charging and discharging heat at a constant temperature during phase change process [1-2]. PCMs are usually classified as non-organic, organic and eutectics kinds [3].

Non-organic PCMs

They have high heat of fusion, inflammable and they are obtainable at low costs. Nevertheless, their main drawbacks is that their super cooling problem during phase change process that results in irreversible transition phase [4]. Hydrated salts are the foremost habitually used PCM for thermal energy storage in this group.

Organic PCMs

They keep a high density energy storage while not abundant much super-cooling compared to inorganic PCMs. They're accessible in a large temperature range and that they have a great agreement with different materials. Paraffin wax, fatty acids and carbohydrate are the most frequently used PCMs in this category [5].

Eutectic PCMs

Eutectic is a mixture of two or more compounds with high density. However, their thermo-physical properties are restricted and the usage of such materials is new for energy storage applications.

The main criterion to select a PCM for a specific application is its phase change temperature. Additionally, different many factors should even be taken under consideration for an appropriate decision: physical properties, non-toxic, large latent heat of fusion, chemically stable, non-erosive. Among these materials, organic PCMs are considered as a kind of vital materials and have an intensive potential for sensible applications in solar energy storage, green house construction and dissipate heat recuperation from electronic devices etc. In spite of their fascinating properties, organic PCMs have some drawbacks like:

- (i) The low thermal conductivity which reduce the rate of heat storage and energy conversion.
- (ii) The low electrical conductivity that frustrates taking advantage of electricity as energy source.

Many methods have been suggested to enhance the electrical and thermal conductivity of pure paraffin, such as inserting fins [6], metal foam [7], dispersing conductive particle within PCM [8] and microencapsulation of PCM [9]. Recently, with the development of nanotechnology, combining pure PCMs with high conductive

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nanoparticles could be crucial to overcome these problems [10]. These nanoparticles such as carbon nanotubes, carbon nanofibers, graphite, metallic nanoparticles, graphene flakes, etc. [11-15] have gained considerable attention in many emerging applications like antistatic coatings, organic light emitting diodes, sensors, flexible microelectronics, and touch screens due to their excellent properties [16-21].

Several numerical and experimental researches are investigated to improve the low thermal conductivity of the PCMs by adding nanoparticles [22-27]. Kant et al. [28] examined the influence of inserting Graphene Nano Particles (GNP) with different concentrations of nanoparticles in various PCM ($\text{CaCl}_2 \cdot 6\text{H}_2\text{O}$, Capric Acid and paraffin). They numerically investigated the melting of these composite materials inside a square cavity heated from one side. They revealed that inserting GNP improves the effective thermal properties of the PCMs and ameliorate the phase change rate. Arasu et al. [29] numerically investigated the melting of nanocomposite (PCM/ Al_2O_3 nanoparticles) with different weight ratio which was heated from two different arrangements. The study described that the melting rate decreases with the heighten of the volumetric composition of alumina (Al_2O_3), the simulation reported that the accumulation of alumina in paraffin wax results in the enhancement of the effective thermal properties for efficient utilization of TESS. Wang et al. [30] prepared paraffin/nano- SiO_2 composites by natural infiltration method as innovative insulation materials applied to electronic thermal protection. Examinations by methods of Thermos Gravimetric Analysis (TGA), Scanning Electron Micrographs (SEM), hot disk analyzer, Differential Scanning Calorimeter (DSC) and thermal protection performance tests were dedicated to the thermal properties, morphology and thermal protection performance analysis of composites. Experimental procedures reported that the incorporation of about 75 wt.% of nano SiO_2 can mainly enhance the thermal properties of the nanocomposite. These excellent properties make the paraffin-nano SiO_2 composite promising for using in the field of electronic thermal protection. Shirazi et al. [31] investigated the feasibility to use PCM nanocomposites for suitable heat management of a Li-ion battery package. The heat generation was modeled using 3D heat transfer model with different nanocomposite structures created by the adding of carbon nanotubes, graphene, and fullerene throughout battery charging/discharging cycles. These results discovered a motivating decrease in temperature variations by embedding the batteries within a paraffin nanocomposite and enhancement of effective thermal conductivity.

Among the research on PCMs focused on electric-to-thermal energy conversion, Zhang et al. [32] elaborated a nanocomposite material by associating paraffin wax as thermal storage matrix and carbon nanofiber as conductive fillers. Then, they evaluated their Electrical Thermal Storage (ETS) performance through an experiment of electric heating and thermal energy release. Results show that electrical resistivity of carbon nanofiber/paraffin wax composites decreases with increasing of the Carbon Nano Fiber (CNF) concentration and the thermal storage capacity is five times better than the traditional thermal storage medium.

To date, most of the research studies focussed on investigating the thermal behavior problem of the phase change material with different volume fraction of nanoparticles. Although, relevant literature related to the electrical behavior are lacking, despite, it plays a vital role in electronic applications such as converting off-peak electricity, use in hybrid vehicles and portable electronics. Thus this is the essential motivation of the current work. Moreover, Graphene is a two

dimensional (2D) honeycomb-like lattice sheet of pure carbon which have excellent electrical [33], mechanical [34], thermal properties [35] and unique structural properties (e.g., lightweight, flexibility) [36]. This combination of proprieties have made a graphene one of the most promising material improvement in different (mechanical, thermal, electrical etc.) properties of produced nanocomposites. In this paper, a numerical study was built to investigate the electrothermal behavior of nanocomposite material paraffin wax filled with graphene nanoparticles. Our challenge is to develop a smart material used in variety of electrical applications such as cooling systems, energy conversion, electronic components, electrodes for rechargeable batteries. Indeed, that can change one or more properties when an electric current is applied and return to this original properties when the stimulus is withdrawn. On this subject, the simulation results will be accustomed to provide a general perspective to conduct experiments that are excessive and time intense as well. The randomly dispersed graphene in the paraffin matrix was simulated to optimized the Electrical Percolation Threshold (EPT). The effects of percolation, concentration level of graphene and temperature on the electrical properties of the nanocomposite material was considered. To understand the physics-based mechanisms during current distribution, a multi-scale numerical models were developed, where the homogeneous mode is used in macroscopic scale and heterogeneous one in microscopic scale.

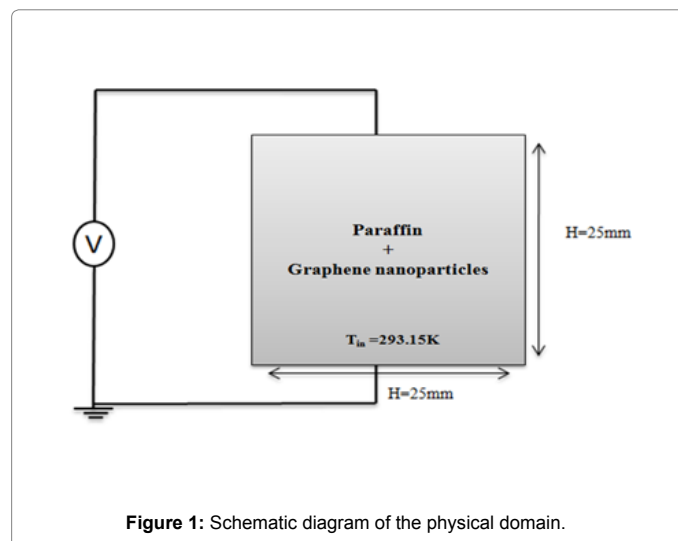
Physical and Numerical Models

Physical model

The dimensions and geometry used for the present study are shown in Figure 1. The graphene nanoparticles with different volume ratios dispersed PCM (nano-PCM) is filled in a square enclosure of size 25 mm×25 mm. Initially, the material is considered as frozen substance (solid). For the electrical boundary conditions, a voltage is applied between the two contact boundaries (top and bottom) of the composite and the two others boundaries are insulated. For the thermal boundary conditions, the four walls are taken at a constant temperature (300 K). The initial velocity of the nano-PCM is zero and maintained no slip at the walls. The initial pressure is at atmospheric pressure. The thermo-physical and electrical properties of PCM and graphene are represented.

The assumptions for the present study are as following:

- 1) The graphene nanoparticles are homogeneously distributed in the PCM.



- 2) The melting of nano-PCM is Newtonian and incompressible, and its flow is two dimensional and unsteady.
- 3) The viscous dissipation terms, volume expansion during melting are negligible.
- 4) The physical properties of the nano-PCM are temperature dependent.
- 5) The melting of nano-PCM is controlled by the convection and conduction modes of heat transfer.

The physics interface solves a current conservation equation for the electric potential leading to the following formulation:

$$\nabla \cdot \vec{J} = \nabla \cdot (\sigma_{NPCM} \vec{E}) = \nabla \cdot (-\sigma_{NPCM} \nabla V) = 0 \quad (1)$$

Where, J is the current density, E is the electric field, V is the electric potential and σ_{NPCM} is the electric conductivity of the nanocomposite.

Then, the temperature distribution is determined by the heat transfer diffusion Eq. (2)

$$\rho C_p \frac{\partial T}{\partial t} + \rho C_p \vec{u} \cdot \nabla T = -\nabla \cdot (-k \nabla T) + S \quad (2)$$

Where, S is a source term present the joule heating due to the circulate of electric current through the material.

$$S = \frac{1}{\sigma_{NPCM}} |\vec{J}|^2 = \sigma_{NPCM} |\vec{\nabla} V|^2 \quad (3)$$

The nanocomposite considered in this study are made of graphene nanoparticles homogenously distributed in the simulation domain during the melting process when the transport of nanoparticles and their rejection by the advancing liquid–solid interface are ignored. In fact the effect of Graphene Nanoparticles (GNP) are considered, the physical properties of such nanocomposites for varying volume fractions need to be properly calculated. The following equation is used for conversion between the mass and volume fractions of nanoparticles:

$$\phi = \frac{\phi_{wt} \rho_{PCM}}{\phi_{wt} \rho_{PCM} + (1 - \phi_{wt}) \rho_{GNP}} \quad (4)$$

Where, ρ_{PCM} and ρ_{GNP} are respectively the density of PCM and the density of graphene nanoparticles.

The density of PCM can be written as:

$$\rho_{PCM}(T) = \rho_{solid} + (\rho_{liquid} - \rho_{solid})B(T) \quad (5)$$

Where, ρ_{liquid} and ρ_{solid} describe the density of the PCM at its liquid and solid phase respectively. The function B(T) as mentioned in Eqn (6), is defined as the liquid fraction was updated to account for the change in specific heat between the solid and the liquid phase of the PCM. The value of function B is 0 when the PCM is totally in solid state and 1 when it became fully liquid. B(T) linearly increases from 0 to 1 between the two states of PCM.

$$B(T) = \begin{cases} 0, & T < (T_m - \Delta T) \\ (T - T_m + \Delta T)/(2\Delta T), & (T_m - \Delta T) \leq T < (T_m + \Delta T) \\ 1, & T > (T_m + \Delta T) \end{cases} \quad (6)$$

Where, T_m is the melting temperature of PCM and ΔT is the

transition temperature of PCM. The liquid fraction of the PCM is used to model the modifications of the thermo-physical properties that appears during the phase transition.

The thermophysical properties of the nano-PCM such as density, specific heat, latent heat, thermal expansion coefficient are obtained with following equations similar to the procedure reported in [26]:

$$\rho_{NPCM} = (1 - \phi)\rho_{PCM} + \phi\rho_{GNP} \quad (7)$$

The modified heat capacity of Nano-PCM is given by:

$$Cp_{NPCM} = (1 - \phi)Cp_{PCM} + \phi Cp_{GNP} \quad (8)$$

Where, Cp_{GNP} and Cp_{PCM} are respectively the density and the heat Capacity of the graphene nanoparticles.

The heat Capacity of PCM is written as:

$$Cp_{PCM}(T) = Cp_{solid} + (Cp_{liquid} - Cp_{solid})B(T) + L_f D(T) \quad (9)$$

Where, Cp_{solid} and Cp_{liquid} are the heat capacity of the PCM at its solid and liquid phases respectively. D(T) is a Dirac delta function and its main role is to distribute the latent heat equally around the mean melting point of PCM. It is set to 0 all over excluding the interval $[T_m - \Delta T, T_m + \Delta T]$ and can be modeled as follows:

$$D(T) = e^{\left(\frac{-(T-T_m)^2}{\Delta T^2} / \sqrt{\pi \cdot \Delta T^2}\right)} \quad (10)$$

When the PCM started to change its phase, it absorbs the latent heat (L_f) which can be represent as a change in its specific heat during the phase transition period.

The dynamic viscosity of the nano-PCM is evaluated by Krieger-Dougherty model for nanoparticles [29].

$$\mu_{NPCM} = \frac{\mu_{PCM}}{(1 - \phi)^{2.5}} \quad (11)$$

The thermal conductivity of the PCM nanocomposite is intended using a Maxwell-Garnett type Effective Medium Theory (EMT) considering the role of thermal boundary resistance between the GNP and surrounding PCM matrix. This model has proven to predict the thermal conductivity enhancement of nanocomposites with reasonable accuracy [23,24].

$$k_{NPCM} = k_d + k_0 \quad (12)$$

Where, k_0 is the stagnant thermal conductivity given by:

$$k_{PCM}(T) = k_{solid} + (k_{liquid} - k_{solid})B(T) \quad (13)$$

The thermal conductivity of PCM $k_{PCM}(T)$ is written as :

$$k_{PCM}(T) = k_{solid} + (k_{liquid} - k_{solid})B(T) \quad (14)$$

Moreover, k_{liquid} and k_d are the thermal conductivity of the PCM at its liquid and solid phases respectively.

The k_d accounts for Brownian motion, which causes the temperature dependence of the thermal conductivity.

$$k_d = 5 \times 10^4 \beta_k \xi \phi (\rho C p)_{PCM} \sqrt{\frac{B_0 T}{\rho_n d_p}} f(T, \phi) \quad (15)$$

Where, B_0 is Boltzmann constant, $1.381 \times 10^{-23} \text{ J/K}$ and d_p is the diameter of nanoparticles.

$$\text{The value of } \beta_k = 8.4407 (100\phi)^{-1.07304}.$$

The value of correction factor $f(T, \phi)$ in the Brownian motion term is defined as the same as

for liquid fraction, $B(T)$ in Eq. (6).

$$f(T, \phi) = (2.8217 \times 10^{-2} \phi + 3.917 \times 10^{-3}) \frac{T}{T_{ref}} + (-3.0669 \times 10^{-2} \phi - 3.91123 \times 10^{-3}) \quad (16)$$

We appropriated that the PCM in the liquid phase is a Newtonian fluid. The mass, momentum and energy conservation equations were resolved simultaneously with the heat transfer diffusion equation. The velocity fields in Eq.(2) is given by Navier-stokes equations for incompressible fluid. However, to model the phase transition, the momentum conservation equation was modified as follow:

$$\rho \frac{\partial \vec{u}}{\partial t} + \rho \vec{u} \cdot (\nabla \vec{u}) - \mu \nabla^2 \vec{u} = -\nabla P + \vec{F}_b + \vec{F}_a \quad (17)$$

Where, the force \vec{F}_b is a buoyancy force can be given by the Boussinesq approximation:

$$\vec{F}_b = -\rho_{liquid} (1 - \beta(T - T_m)) \vec{g} \quad (18)$$

And the force \vec{F}_a is an additional source to damp the velocity in solid PCM

$$\vec{F}_a = -A(T) \cdot \vec{u} \quad (19)$$

With the function $A(T)$ inspired from the Carman-Koseny relation for porous medium

$$A(T) = \frac{C(1 - B(T))^2}{(q + B^3(T))} \quad (20)$$

Where, q is typically a small number so as to make Eqn (20) effective, even when the liquid fraction $B(T)$ is zero. The constant value of q was fixed at 10^{-3} . C defines the mushy zone constant and its value is subject to the morphology of the PCM. The mushy zone constant describes how steeply the velocity is totally reduced to zero when the PCM becomes completely solid. In this research, C is taken as a constant value 10^6 [28]. This value is chosen arbitrary high due to the high viscosity of solid PCM.

Numerical model

COMSOL Multiphysics software is used to solve numerically the concurrent equations mentioned previously in Section (2.1) and subjected to the boundary conditions based on the finite element method. The physical properties of Nano-PCM are defined according to temperature, which determines the phase (liquid, mushy, or solid) and nanoparticles concentration (ϕ). The numerical simulation is carried out according to the following steps: Discretization of the considered domain (size, element and type), defining the appropriate time step as well as the relative and absolute tolerances and use the appropriate solver techniques. In order to get the most suitable solution, a mesh dependence study was performed carefully for the current investigation. A triangular meshing with a regular refinement method was used for the global mesh at the macroscale geometry of the

nanocomposite material as shown in Figure 2a. Firstly, 1954 domain elements are considered for the entire model calibrate for the fluid dynamic with the predefined normal mesh. The maximum element size and element growth rate are set to 0.00113 and 1.15 respectively. Secondly, 4412 domain elements are considered for the entire model calibrate for the fluid dynamic with the predefined fine mesh as shown in Figure 2b where the maximum element size and element growth rate are set to $7E-4$ and 1.1 respectively. Finally, 17800 domain elements are considered for the entire model calibrate for the fluid dynamic with the predefined extra fine mesh. The maximum element size and element growth rate are set to $3.25E-4$ and 1.08 respectively.

It is well noted that there is not a significant difference in the obtained results for all mesh sizes. For better precision and exactness, a mesh size with 17800 elements will be chosen for our research. An absolute tolerance of 0.00001 and a relative tolerance of 0.001 are applied. The time stepping is selected with an initial time step of 0.01 s and a maximum time step of 100 s.

Then, in order to simulate the electrothermal behavior of the PCM nanocomposite, a coupling of two different COMSOL physics interfaces was required. We therefore used the electric current and heat transfer models which the interrelationship of the way they interact with each other is shown in Figure 3. The electric currents interface applies a voltage between the two contact boundaries of the nanocomposite. The overall process of simulation is to control the change of temperature inside the material when an electric current is applied. In order to couple the electric currents interface from with the heat transfer interface, the power dissipation density is used as input heat source defined by the joule heating. The velocity and pressure distributions are also used as

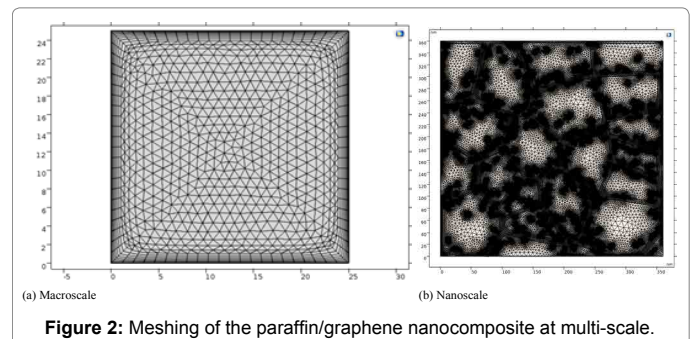


Figure 2: Meshing of the paraffin/graphene nanocomposite at multi-scale.

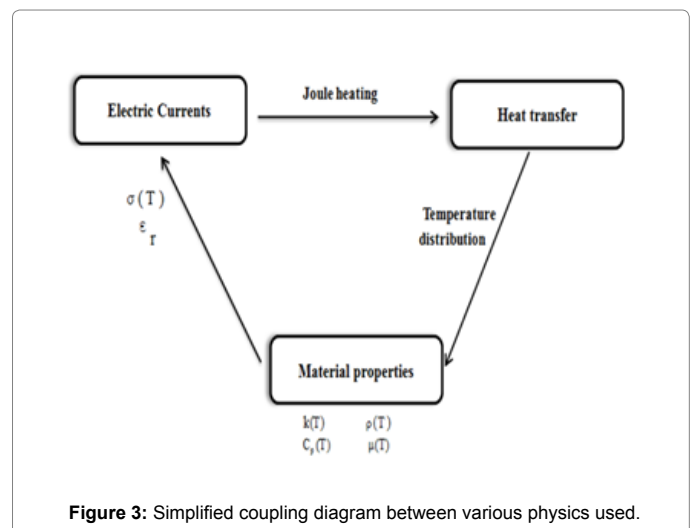


Figure 3: Simplified coupling diagram between various physics used.

inputs in the heat transfer interface.

Stochastic microstructure

Our goal is therefore to develop techniques to generate the nanocomposite material (GNP/paraffin wax) with well-controlled structural properties. The proposed method includes a stochastic Voronoi diagram to control the nanoparticle size and distribution, and a B-spline curve to control the particle shape [37,38]. In the literature, Voronoi tessellations have been widely used to divide geometric domains and generate polygonal meshes [39,40].

The characteristics of this structure are modeled through some functions to optimize the topology of nanostructure with the aim to define the physical parameters such as volume fractions. Different steps were developed to use the appropriate functions, as detailed in the organizational chart in Figure 4.

The first step of structural modeling is to determine the random generation of different points $P = \{p_1, p_2, \dots, p_n\}, n \geq 2$ in the m -dimensional space R^m shown in Figure 5a. These points will be connected together to create irregular convex and concave polygons where the assembly of these polygons form the Voronoi cell of p_i which follows Eq. (21)

$$V(p_i) = \{x \mid \|x - x_i\| \leq \|x - x_j\|, \text{ for } i \in I_n, j \in I_n, j \neq i\} \quad (21)$$

Where, $\|x_1 - x_2\|$ is the Euclidean-distance between two point x_2 and $x_1, I_n = \{1, 2, \dots, n\}$.

In R2, the boundaries of Voronoi cells contain line segments that connect exactly perpendicular bisector between two generators points. The boundary is called Voronoi edge, and all the Voronoi edges of the same Voronoi cell form a Voronoi Polygon. The point where three or more Voronoi edges meet is called Voronoi point or Voronoi Vertex shown in Figure 5b.

Let $V(p_i)$ a voronoi cell associated with the generator $V(p_i)$ and the vertices of the voronoi polygon $V(p_i)$ be $\{v_0, v_1, \dots, v_m\}$ which are modeled as control points of closed B-Spline curves $C(u)$ of degree k , we take

$$C(u) = \sum_{i=0}^m N_{i,k}(u) v_i \quad (22)$$

Where, $N_{i,k}(u)$ is B-Spline basis function of degree k , which can be recursively defined by Eqn (23). To create a periodic B-Spline curve, the first and last k control points must be properly wrapped in the form such that $v_{k-1} = v_m$.

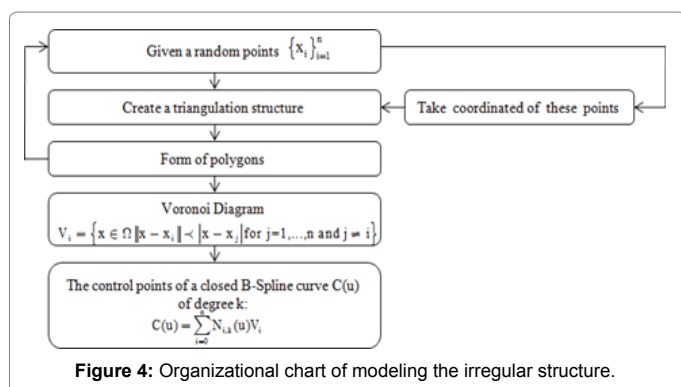


Figure 4: Organizational chart of modeling the irregular structure.

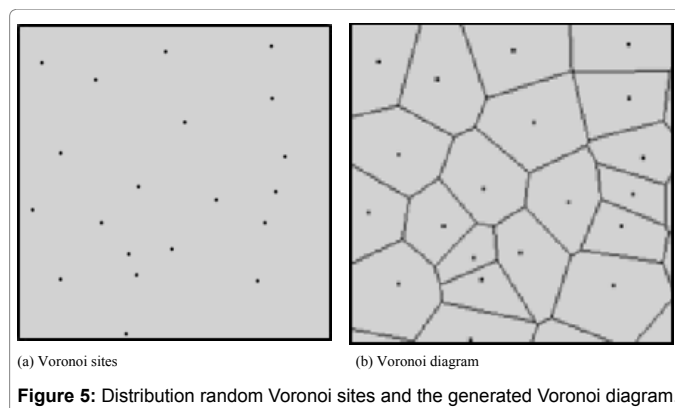


Figure 5: Distribution random Voronoi sites and the generated Voronoi diagram.

$$N_{i,0}(u) = \begin{cases} 1 & (u_i \leq u < u_{i+1}) \\ 0 & (\text{otherwise}) \end{cases} \quad (23)$$

$$N_{i,k}(u) = \frac{u - u_i}{u_{i+k} - u_i} N_{i,k-1}(u) + \frac{u_{i+k+1} - u}{u_{i+k+1} - u_{i+1}} N_{i+1,k-1}(u) \quad (24)$$

The B-Spline curves are then used to allow for a closer fit of observed structures with irregular cell size and shape.

This allows for a more realistic simulation of the properties of the materials. Furthermore, the generation of virtual nanomaterial is possible which can then be used for optimizing the existing microstructure and for the virtual design of new materials shown in Figure 6.

Indeed, before presenting the B-spline curves, the merged polygons are scaled down with a constanscaling factors ($0 < s < 1$) to scale all the B-spline curves and modify the thickness of interstices.

Model Validation

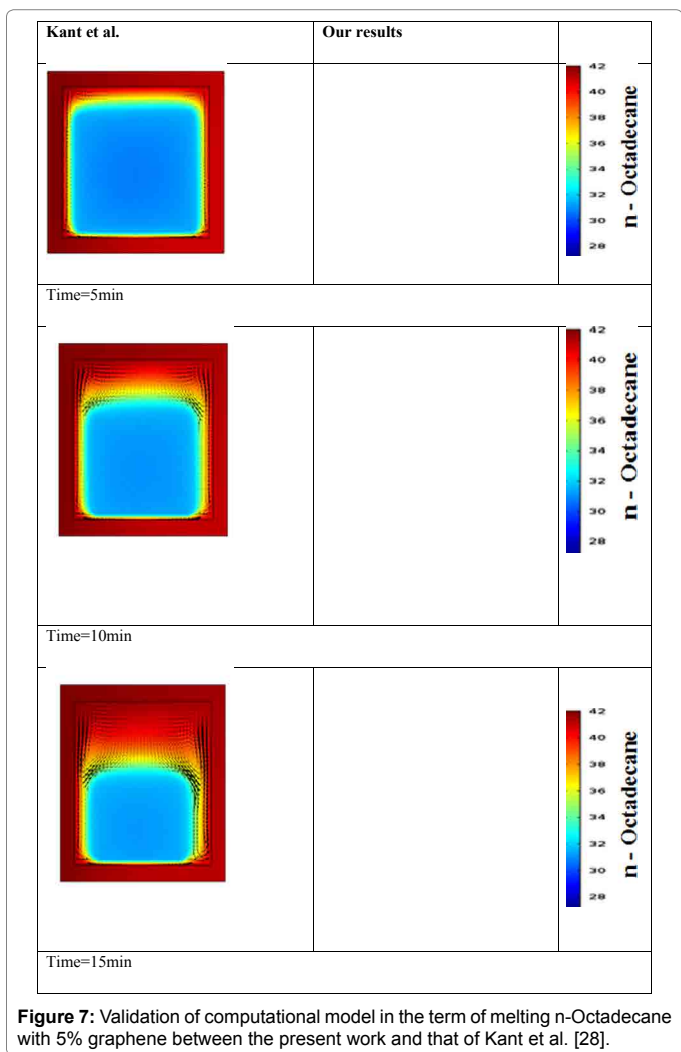
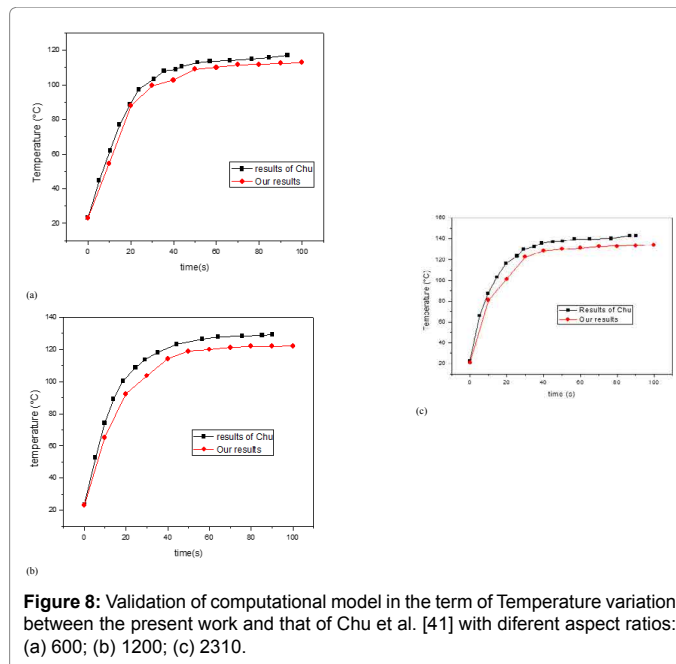
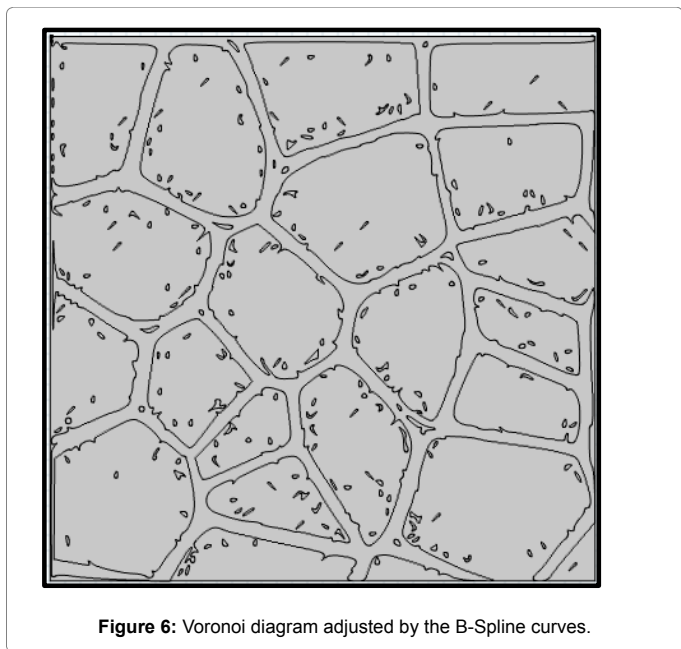
This section presents a validation approach and illustrates the accuracy of the computational model by comparing with different research. In first step, the calculation from the current simulation model, in terms of thermal behavior, are compared with the numerical results of Kant et al. [28] and shown in Figure 7. Initial computational were achieved using the developed methodology of the present investigation for which similar structure geometry, boundary conditions and the same material properties were used as designated by Kant et al. [28]. As show in Figure 7, the temperature distriution and the velocity field of the PCM nanocomposite are in good agreement with the results of numerical study carried out by Kant et al. [28]. Such comparative study establishes a stringent test of the agreement of our thermal model in the term of melting nano-PCM and its predictions before using it further.

In second step, to validate our proposed electrothermal behavior, we initiated by modeling a previously published work of Chu et al. [41]. The electric heating properties of the CNT/PDMS composites (5.6%) with different aspect ratios were evaluated by measuring the changes in the temperature under an applied DC voltage of 10V. This comparative study clearly shows in Figure 8 that the developed electrothermal model is in good agreement with the results of the experimental study carried out by Chu et al. [41].

Results and Discussion

Electrical-thermal behavior

The Figure 9a shows the electrical potential distribution and the



current density field product from electrical analysis. The red color indicates the highest voltage of 23 V which was set in the boundary condition. The opposite side has the zero electrical potential. When we apply an electric potential to the nanocomposite material (paraffin wax/4wt.% of GNP), an electric current circulate through the material defined by a current density vector per cross-sectional area, its direction being that of the motion of the charges. By applying a given DC voltage, the material can generate Joule heat proportional to the resistance and square of the current. Figure 9b shows the distribution of the temperature inside the material. It illustrates that a voltage of 23 V increases the temperature from room temperature to 320 K in 300 s, as the maximum value was detected in the center of the media. The solid nano PCM absorbs heat and transforms into a liquid state, produces a self melting. The velocity vector near the solid nano PCM is high due to a higher temperature gradient. Infact, energy from the electrical field is converted into thermal energy.

Effect of GNP concentration

Electrical conductivity: The primary goal of this simulation is to calculate the effective electrical conductivity of the composite medium made of paraffin wax and graphene nanoparticle. Figure 10 shows the electrical conductivity of the composite with different GNP weight percentages at fixed voltage of 23V. The unfilled neat paraffin (at 0 wt.%) is an insulator that exhibits low electrical conductivity of 1.6×10^{-14} S/m. When GNP nanoparticles were added to paraffin, the electrical conductivity increased. On the other hand, electrical conductivity of the composite with low GNP concentrations level of 0.3-2 wt.% was measured to be around of 1.6×10^{-12} - 7.3×10^{-11} S/m, which was quite similar with that of the pure paraffin. It means that the electric charges do not flow effectively in those composite because of no physical connection between graphene sheets dispersed in the paraffin matrix. On the other hand, for the composite with high graphene contents of 2.5-4.5 wt.%, the electrical conductivity was dramatically increased to be around of 1-3 S/m. Therefore, it is reasonable to note that the electrical percolation of nanocomposite is related to the concentration of fillers inside the matrix because conductive networks formed at certain graphene content between 2.0 and 2.5 wt.%. To quantitatively

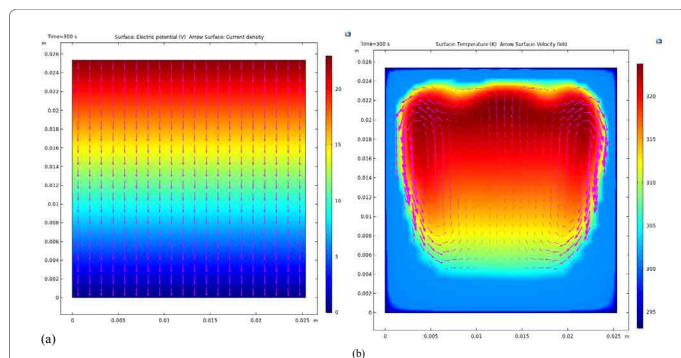


Figure 9: The distribution of electric potential and temperature after 300s electric heating in the composite with 4 wt.% graphene contents at 23 V applied voltage.

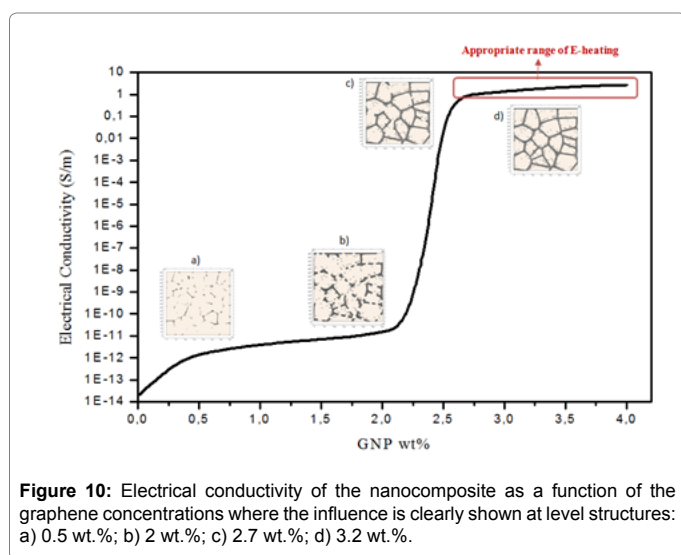


Figure 10: Electrical conductivity of the nanocomposite as a function of the graphene concentrations where the influence is clearly shown at level structures: a) 0.5 wt.%; b) 2 wt.%; c) 2.7 wt.%; d) 3.2 wt.%.

understand the abrupt change in the electrical conductivity, we have considered the material (PCM/graphene) at nanoscale by setting the domain (350 nm×350 nm). We proposed a model that traduce realistic virtual structure of the nanocomposite through the optimization of embedded graphene nanoparticles: shape, volume fraction, size distribution, and spatial arrangement. This model is based on Voronoi function and B-Spline curves as shown in Figure 6. At low GNP content, the electrical conductivity of the GNP/paraffin nanocomposites was very close to the conductivity of the paraffin matrix, consequently of no conductive GNP network was formed as shown in Figure 10. It means that the electric charges do not flow effectively in these nanocomposites because of the vast distance between the GNP which make difficulty to form the conductive path inside the paraffin matrix. The increase of GNP leads to a reduction in the average space between neighbors GNP added in composites and the electronic transition by tunneling conduction becomes easier. The alignment of the nanoparticles crisply increases the electrical conductivity of the nanocomposites along the direction of the electrical field. As more GNP were added, the nanoparticles moved close together, and at a certain concentration (percolation threshold), a continuous conductive network of graphene nanoparticles was formed and the conductivity increased by several orders of magnitude.

Figure 10d shows a close view for 3.2 wt.% GNP dispersed in the paraffin forming network exhibited percolation behavior. It illustrates the conductive network which occurs as the GNP are getting close

to each other, creating electron paths through the matrix and thus increasing electrical conductivity. The conductive behaviour in these three regimes can be understood from the nanostructure of the composite. When there is no conductive path through the filler, no charge can flow, and the composite remains insulating. On the other hand, if the filler forms a directly connected network, electrons can move through this network, and the composite is conductive. To achieve a highly conducting composite, the intrinsic poor conductivity of paraffin wax can be compensated through the use of the conductive GNP that form a conducting network.

Thermal conductivity: The thermal conductivity of the nano PCM with different graphene concentrations is calculated and represented in Figure 11. The effective thermal conductivity of the composite PCM increases with increasing of graphene contents. The high percentages are involved with a high volume of graphene nanoparticles which results in an increase in the conductive paths for heat flow.

Velocity field: The velocity magnitude distribution and streamline plot of the nanocomposite with different concentration of GNP are figured out in Figure 12. At the onset of melting process, the phase change of the nanocomposite material from solid to liquid phase starts from the center. As the density of PCM reduces, the nanocomposite is melted and transfer upward transferred due to the presence of the buoyancy force. At nanocomposite with 3 wt.% of graphene contents, the velocity streamlines form four snapshots inside the media (Figure 12a) and the symmetry of the circulation pattern is still preserved. At the right side, the direction of circulation is clockwise and anti-clockwise at the left side. During fusion process, two identical circulations (eddies) of velocity streamlines inside the composite at 4 wt.% of graphene contents appear as shown in Figure 12b. Moreover, the vortices are symmetric and a small distance between the two vortex centers is observed.

Voltage effect

Electric heating experiments of the graphene/paraffin composite with different graphene contents were carried out by varying the applied voltage in the range of 5-40 V. When an external voltage of 30 V was applied to the composite, the temperature was momentarily increased. By observation the temperature of the composite at an applied voltage with the time, the electrical heating behavior was investigated. Figure 13 displays time-dependent temperature changes of the composite

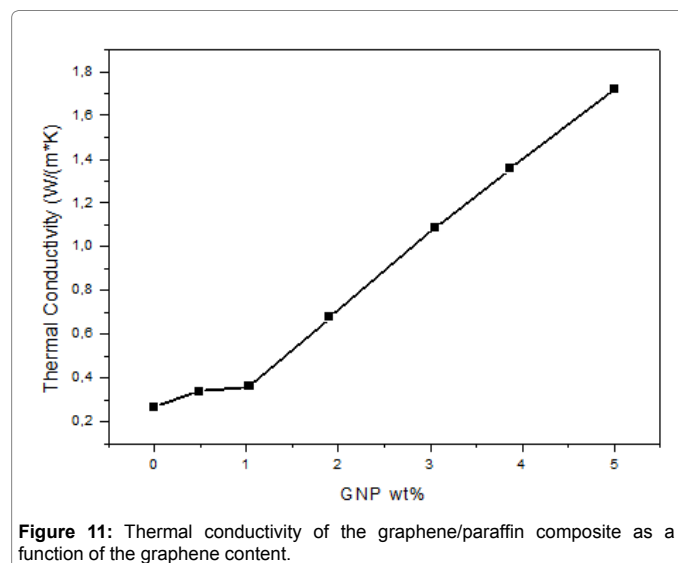
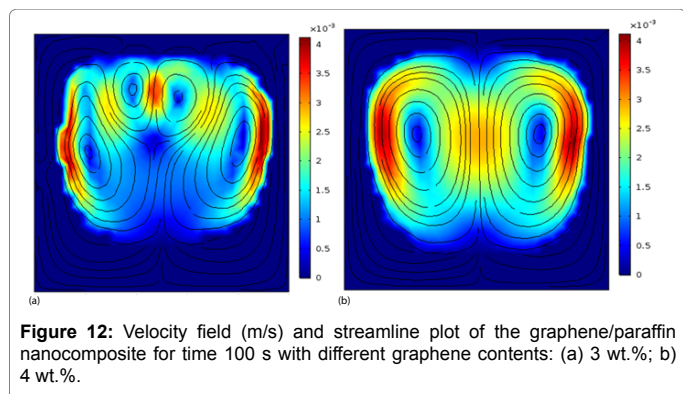


Figure 11: Thermal conductivity of the graphene/paraffin composite as a function of the graphene content.



a considerably increase in temperature of the studied nanocomposite material was observed with time due to the application of voltages above 20 V. Then, maximum temperatures were attained within 300 s and maintained constant over time.

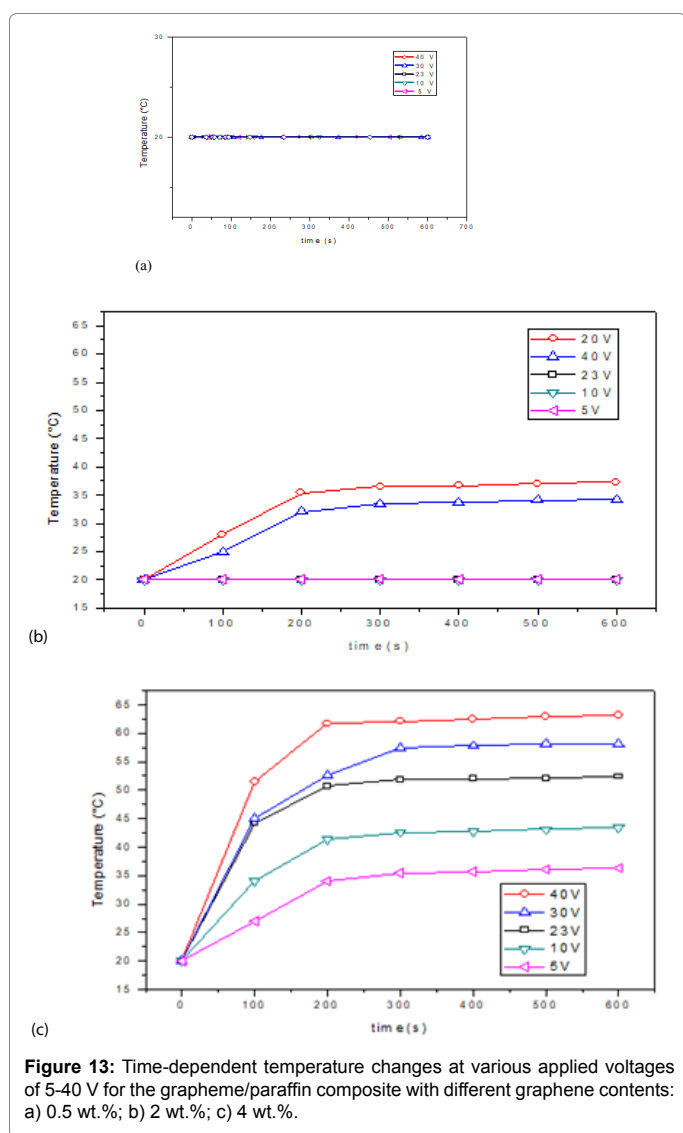
This electric heating behavior was similarly monitored for other composite with higher graphene contents of 3.0-4.5 wt.%, except for the fact that the great temperature attained at a given applied voltage was higher for the nanocomposite with higher graphene contents (Figure 13c). So the nanocomposites with higher GNP weight could have longer electrical network path, resulting a higher electrical conductivities. Therefore, much more current flows resulting in higher temperature. Thanks to the high thermal conductivity of the GNP nanoparticles, the saturated temperature is easily reached in a faster and easier way while increasing voltage.

Conclusion

In this study, a Finite Element Method (FEM) were investigated for solving the energy, electric and dynamics equations in the composite material (paraffin wax/graphene nanoparticles).

Based on the numerical results, we proved that:

- The effective electrical conductivity of the nanocomposite increases sharply with the concentration of graphene nanoparticles.
- The random dispersion of graphene nanoparticles inside paraffin wax matrix predicts an Electrical Percolation Threshold (EPT) around of 2.5wt.%. Below this value of EPT, electrical conductivities were around of about 3.5×10^{-12} to 7.3×10^{-11} S/m, which was quite similar with that of the pure paraffin i.e 1.6×10^{-14} S/m because the absence of connected pathways in the materials. In this state, Graphene/paraffin wax nanocomposites are suitable materials for applications requiring insulation that has a low electrical conductivity and important thermal properties.
- Above the same value of EPT, the well-dispersed electrically conductive GNP networks were found to act as nanoheaters integrated in the paraffin matrix through simple Joule heating when an electric current was passed through them. In this state, Graphene/paraffin wax nanocomposites are suitable materials for electric heating applications, having appropriate thermal properties.
- The expansion of paraffin wax is the most important factor for the decreasing of electrical conductivity in the composite GNP/paraffin wax.
- The increase of applied voltage involved an increase of the nanocomposite temperature at high concentration level of graphene (above EPT) nevertheless without effect at low concentration (below EPT).
- The effective thermal conductivity of the nanocomposite increases linearly with the concentration of graphene nanoparticles.



with different graphene contents at various applied voltages. For the nano-PCM with 0.5 wt.% graphene content, there was no temperature change over the applied voltage range of 5-40V (Figure 13a), which demonstrates that there was no electric heating behavior. In case of the nano-PCM with 2.0 wt.% graphene, the temperature was not changed with time at relatively low applied voltages of 5-23 V (Figure 13b). But,

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